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Synthesis and evaluation of ATO as a support for $Pt-IrO₂$ in a unitized regenerative fuel cell

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abstract

An IrO₂ catalyst was prepared using a colloidal method followed by a thermal treatment. The catalyst was later mixed with Pt-Black and supported on the Sb-doped SnO₂ (ATO), synthesized through the same colloidal method. ATO was investigated as a possible catalyst support in an electrode of a regenerative fuel cell (URFC), where Pt-IrO₂ was used as the catalyst for the oxygen evolution and reduction reactions. The morphology and composition of the ATO support was investigated through transmission electron microscopy, X-ray diffraction (including Rietveld Refinement), BET analysis, and X-ray fluorescence. An ATO support was obtained with a highly homogeneous distribution and crystal sizes, measuring approximately $4-6$ nm.

The Pt-IrO₂/ATO material was deposited on a Nafion 115 membrane with 0.5 mg cm⁻² of catalyst loading. Pt/Vulcan XC-72 (30 wt. %, E-TEK) was used as the catalyst in the H_2 compartment with a Pt loading of 0.4 mg cm $^{-2}$. The electrochemical activity of the Pt–IrO₂/ ATO for oxygen evolution/reduction in the URFC system was investigated by ACimpedance spectroscopy, linear voltammetry, and chronoamperometry techniques. The maximum mass current activity was 1118 A g^{-1} at 1.8 V in proton-exchange membrane water electrolyser mode (PEMWE) and 565 A g^{-1} at 0.3 V in proton-exchange membrane fuel cell mode (PEMFC), both at 80 °C. The value of the round-trip energy efficiency was approximately 48% at 50 A g^{-1} .

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1. Introduction

Electrolysers and fuel cells are the most promising technologies for production and utilization of hydrogen as an energy storage medium $[1-9]$. Both systems may alternatively operate in the same device based on a proton-exchange membrane (PEM). The combination of both processes in the same system is known as unitized regenerative fuel cell (PEM-URFC). URFC devices are characterized by a fast startup, efficient operation at low temperature without significant power

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consumption and stable operation, even after a large number of startup/shutdown cycles.

During hydrogen production by water electrolysis in PEM system (PEMWE), most of the overpotential responsible for losses is related to the electrochemical processes at the anode, where the O_2 evolution reaction takes place [10]. Ru O_2 is considered the most efficient catalyst for the oxygen evolution reaction (OER) in the alkaline electrolyte $[11-13]$; however, in acidic media, $RuO₂$ undergoes strong degradation during $O₂$ evolution with a consequent loss of its chemical reactivity. $RuO₂$ can form different sub-oxides that affect its stability and the registered overpotentials [14]. Certain materials, such as IrO₂, are used as catalysts due to their similar properties to those of $RuO₂$. It is even possible to use chemical mixtures of both oxides at different ratios [2,15,16].

In fuel cell mode (PEMFC), the limiting process occurs at the cathode, where the oxygen reduction reaction (ORR) takes place on metal electrodes, such as Pt, Pd, Ru, Ir, Rh, Au, and alloys [17,18] in acidic environments.

One of the challenges for the PEM-URFC is the development of new catalyst supports that can reduce the catalytic loading of the materials and increase their stability in PEMWE systems, where carbon-based materials are subjected to high potentials, humidity and temperatures that will accelerate the corrosion process. Such characteristics as high surface area, good electrical conductivity, and high resistance to corrosion at a range of potentials between 0 and 2 V are required for a catalyst support with this potential application.

A number of materials with attractive features as catalyst support have already been used, such as $M-TiO₂$, WO₃, S-ZrO₂, Sn-In₂O₃, WC, and Ti_nO_{2n-1} [19-30]. The most common for OER is the Ebonex® (Atraverda Ltd., UK), which consists of an electrically conductive ceramic material based on titanium sub-oxides, mainly $Ti₄O₇$ and $Ti₅O₉$. These suboxides are the main conductive compounds in a homologous series of crystallographic shear structures with the general formula Ti_nO_{2n-1} (4 \leq n \leq 10), known as the Magneli phase [30]. Recent studies have shown the potential use of this material as a catalyst support for the OER and the ORR $[31-33]$; however, this material has low surface area (1.7 $\mathrm{m^{2}}\mathrm{\ g^{-1}}$) compared with Vulcan XC-72 (205 m² g⁻¹) [32,34].

Another material that has shown potential as a catalyst support is Sb-doped SnO₂ (ATO), due to its low electrical resistance (10 $^{-2}$ –10 $^{-3}$ Ω cm) compared with Ebonex® (10^{-3} Q cm) [35] and its enhanced surface area. Another unique feature is its stability in acidic media [36]. Recent studies have reported the use of commercial ATO (Sigma-Aldrich) with a surface area of 47 m^2 g^{-1} and average particle size of 30-40 nm, which resulted in a high distribution of catalytic material available for the OER in PEMWE mode [37,38]. This paper presents the synthesis and evaluation of ATO with a surface area value superior to commercial ATO, and compares its performance as a catalyst support for dualfunctions mixed with $Pt-IrO₂$ as a catalyst in a PEM-URFC. ATO was synthesized and characterized by physicochemical and electrochemical techniques as a dual-function catalyst support for PEMFC and PEMWE modes. Conversion efficiencies and performance in the PEM-URFC cell were determined and compared with previously reported results obtained with Ebonex $^{\circledast}$ as the catalyst support.

2. Experimental

2.1. Synthesis of the ATO support

The ATO material was synthesized by dissolving 2.65 g of SnCl4.5H2O (Strem chemical) in 20 mL of HCl, then heating (60 \degree C) and stirring during 30 min. At the same time 0.187 g of $SbCl₃$ (Strem) was dissolved in 2 mL of HCl. Both solutions were mixed during 20 min and subsequently combined with a 1:1 ethanol:isopropanol mixture, under air atmosphere and magnetically stirred for 4 h. The sample was heated ($\leq 75^{\circ}$ C) to obtain a precipitate and then calcined at 500 $\,^{\circ}$ C for 12 h.

2.2. Preparation of Pt $-IrO₂/ATO$

Synthesis procedure of $IrO₂$ has previously been reported [15]. A sample of IrO₂ was mixed mechanically with Pt-black (E-TEK) 50:50 wt.%. Afterwards, the catalysts were supported (in a 30% wt. ratio) on Sb-doped $SnO₂$ (ATO) by sonication, in an ultrasonic bath of isopropanol at 80 \degree C, until a dry powder was obtained. This powder was subsequently crushed and mixed with a 33% Nafion® solution (Electrochem, 5% in isopropanol) for later deposition on a Nafion 115 membrane.

2.3. Physicochemical characterization of ATO

X-ray diffraction (XRD) was performed in a Rigaku DMAX-2200 diffractometer, using CuKa radiation at 40 kV and 40 mA. The diffractometer was operated at 0.08 $^{\circ}$ $\mathrm{min}^{-1}.$ The collected data were refined by the Rietveld method using FULLPROF SUITE software [39,40]. The Thompson-Cox-Hastings pseudo-Voight function was used to model peak shapes and preferred orientation. The background was modeled as a polynomial with 6 refined coefficients. The diffraction patterns were fitted to the JCPDS (Joint Committee on Powder Diffraction Standards) database. The surface area of the ATO was estimated according to the Barrett-Joiner-Halenda (BJH) method, which was then used for the determination of the cumulative area and the pore sizes measured in an Autosorb® IQ Gas Sorption Analyzer (Quantachrome).

X-ray fluorescence analysis (XRF) of the catalysts was carried out in a Bruker AXS S4 Explorer spectrometer, operating at a power of 1 kW and equipped with a Rh X-ray source, a LiF220 crystal analyzer, and a 0.12° divergence collimator. The morphology of the ATO support was investigated with a Phillips TEM microscope Model CM200 operated at 200 kV. The TEM samples were prepared by dispersing the ATO in isopropyl alcohol by ultrasonic treatment dropping the suspension onto carbon-coated grids.

2.4. Preparation of membrane-electrode assembly (MEA)

A Nafion 115 (Ion Power) membrane was used as solid polymer electrolyte. The oxygen reduction/evolution catalyst supported in ATO was directly deposited onto one side of the membrane using a hot-spray technique. Inks were composed of aqueous dispersions of the catalyst, catalyst support, deionised water, Nafion® solution (5% Aldrich), and anhydrous Ethanol (J.T. Baker); $Pt-IrO₂$ (50:50 wt.%) catalyst loading

was 0.5 ${\rm mg}\,{\rm cm}^{-2}$. A commercial 30% Pt/Vulcan XC-72 (E-TEK, PEMEAS, Boston, USA) was used as the catalyst for the dualfunctional hydrogen electrode. The hydrogen electrode catalyst ink was prepared by mixing a suspension of Nafion ionomer in water with the Pt/C catalyst powder [41,42] in an ultrasonic bath. The obtained ink was spread on carbon cloth backings (GDL ELAT from E-TEK) with a Pt loading of 0.4 $\rm{mg \ cm^{-2}}.$ The ionomer content in both electrodes was 33 wt.% in the catalytic layer after drying. The geometrical area of the electrode was 5 $cm²$. The MEA was directly installed in the cell housing and compressed by tightening to 9 N m using a dynamometric wrench.

2.5. Electrochemical characterization of MEA

URFC electrochemical behavior was evaluated by linear voltammetry, chronoamperometry and electrochemical impedance spectroscopy (EIS) in a frequency range of 10 kHz to 1 Hz. The electrochemical characterization was performed with a PGSTAT Autolab 302 Potentiostat/Galvanostat equipped with a booster of 20 A (Metrohm $^\circledast$) and a Frequency Response Analyser (FRA). The fuel cell test station, which operated on liquid fuel (ElectroChem Inc. with a direct methanol unit couple), was used to control operating parameters, such as cell temperature, inlet temperature of gases and liquids, gas humidity, pressure, and flow rate.

The electrolyser performance was evaluated at 80 $^{\circ}$ C. Deionised water was circulated at 2 mL min^{-1} in the oxygen electrode compartment. Linear voltammetry profiles in PEMWE mode were performed for a range of potentials from 1 to 1.8 V at a 50 mV s $^{-1}$ scan rate. The fuel cell performance test was carried out under atmospheric pressure at 80 $^{\circ}$ C by feeding dry H_2 and air, at 50 and 100 scc of flow, respectively. The linear voltammetry curves in PEMFC mode were performed from open circuit voltage (OCV) to 0.3 V with a 50 mV s $^{-1}$ scan rate.

3. Results and discussion

Fig. 1 shows the obtained diffraction pattern of the ATO support with its corresponding Rietveld refinement. From this analysis, it was possible to identify $Sn_{0.90}Sb_{0.1}O_2$ in the rutile phase (JCPDS 88-2348). Presented in this figure are the observed pattern (Yobs), the calculated pattern (Ycalc), the difference between them (Yobs-Ycalc), and their corresponding Bragg positions. Good coherence between the experimental and calculated diffraction pattern were observed.

The mean crystallite size was estimated from the broadening of the main peaks using the Debye-Scherrer equation. Particle dimensions of 4-6 nm were obtained for the ATO support. These sizes are smaller than those reported for commercial ATO supports [37]. To verify the presence of Sb in the support, an XRF study was conducted which yielded the results listed in Table 1. The ratio between Sn and Sb is approximately 10:1. During the XRF testing, chloride ions were also detected, at levels below 1%, these ions can appear as contaminants during the reactions of interest. Even though the synthesized ATO was washed several times, it is not possible to eliminate this contaminant completely.

Fig. 1 – X-ray diffraction pattern of Sb-doped $Sn_{0.9}Sb_{0.1}O_2$ (ATO) synthesized and calcined at 500 \degree C for 12 h, and its corresponding Rietveld refinement plot. Space group P42/ mnm, a = 4.7338 Å, b = 4.7338 Å, c = 3.1928, $\alpha = 90^{\circ}$, $\beta = 90^{\circ}, \gamma = 90^{\circ}, \chi^2 = 0.8, R_{wp} = 0.22, R_p = 0.292,$ $R_{exp} = 0.5076.$

Surface area and pore volume values estimated through the BJH method (c = 0.8) resulted in values of 115.17 m^2g^{-1} and 0.272 $\rm cm^3\, g^{-1}$, respectively, for ATO; while the reported values for Ebonex® are 1.17 $\mathrm{m^2 g^{-1}}$ and 0.01 $\mathrm{cm^3 g^{-1}}$ [43]. The analysis of the pore size distribution ($N₂$ desorption branch) indicated the presence of mesopores in the ATO support. It is remarkable that the surface area value obtained for ATO support is superior to the value of the catalytic material [15]. The ATO support could provide a possible effective method for the dispersion of catalytic materials on the substrate as opposed to current method of commercially available Ebonex®. It is expected that ATO will have a contribution in the performance.

Fig. 2 shows the TEM micrographs; spherical aggregates and networks can be seen. A careful analysis of several micrographs allows the estimation of the mean diameters of the primary particles (Fig. 2B). The average particle size was 4.4 nm. The mean sizes were comparable with the results obtained by XRD.

Fig. 3 shows the EIS spectrum under PEMFC and PEMWE modes at 80 °C and atmospheric pressure. Under these conditions the oxygen reactions represent the determining step and the hydrogen electrode can be assumed to act as pseudoreference electrode [44]. The value of the series resistance (R_s) was 0.27 Ω ·cm² for Pt-IrO₂/ATO in PEMFC mode, whereas the charge transference resistance (R_{ct}) was 4.5 Ω ·cm². In PEMWE mode, the R_s was 0.28 $\Omega \cdot \text{cm}^2$ and the R_{ct} was 2.38 $\Omega \cdot \text{cm}^2$. The R_s values were similar to those obtained from commercial Ebonex® according to Siracusano et al.; however, the R_{ct} value was lower than that reported in PEMWE mode [43].

Fig. $2 - TEM$ micrographs of ATO support: (A) and (B) Particle size distribution analysis.

The performance of the MEA based on the $Pt-IrO₂/ATO$ system for oxygen evolution/reduction reactions was evaluated at 80 °C for both PEMWE and PEMFC (Fig. 4). The PEMWE performance is higher than that obtained when commercial Ebonex $^{\circledast}$ and $\mathrm{Ti_{n}O_{2n-1}}$ were used as the catalyst support [43]. At 1.8 V, these are 300 A $\rm g^{-1}$ and 700 A $\rm g^{-1}$ for IrO $\rm _2$ /Ebonex $^{\circledast}$ and $IrO₂/Ti_nO_{2n-1}$, respectively; meanwhile at same cell voltage a mass activity value of 1118 A $\rm g^{-1}$, corresponding to Pt $-$ IrO $_2$ supported on ATO was achieved. The overpotential at the onset of the OER was 1.42 V versus 1.40 V for Ebonex® [33]. In the PEMFC mode, however, higher activity was observed for the ORR by the support materials than the ATO support; the mass activity was 565 A g^{-1} at 0.3 V and 80 °C. It is possible that the ATO support participates in the OER in PEMWE mode in acid media according to C. Xiao et al. [45], but not for the ORR in PEMFC mode. According to different authors, including Damjanovic, Brusic, and Know [46-50]; this result may be attributed to the contribution of surface oxide formation on Pt,

Fig. 3 – IES spectra of Pt-IrO₂/ATO support in PEMFC mode (0.6 V) and PEMWE mode (1.55 V) at 80 \degree C.

it has been shown that for a potential near 0.8 V versus NHE there is a type of oxygen adsorption on the surface of the Pt (Pt-O). Beside this, Thacker and Hoare [49] demonstrated that for potentials between 1 and 2 V versus NHE, certain $Pt-O$ sites may react to become PtO₂ or, in the presence of $O₂$, and the interaction of the gas with the surface of the Pt may cause Pt $-O₂$ to form. This possible explanation can be attributed to the contribution of the Pt material, due to surface oxides created by the potential, which works in PEMWE, because it is known that PtOx is active for OER [51]. This finding is not observed in PEMFC, since $IrO₂$ and support (ATO) are not active in carrying out the ORR and may block active Pt sites, decreasing the performance of the system in this modality.

Fig. 5 shows chronoamperometric curves from testing, which was carried out at 80 $^{\circ}$ C and atmospheric pressure for the Pt-IrO₂/ATO materials. These experiments were performed by applying a potential of 0.6 V for 400 min in PEMFC mode, and subsequently at 1.55 V for the same amount of time in PEMWE mode. The results showed an acceptable stability for this lapse of time in the two different functions.

Fig. 4 – URFC polarization curves at 80 \degree C created with the MEA based on the $Pt-IrO_2/ATO$ material.

Fig. 5 – Chronoamperometric curves of Pt-IrO₂/ATO material in PEMFC and PEMWE modes at 0.6 V and 1.55 V, respectively, at 80 \degree C.

The round-trip energy conversion efficiencies (\mathcal{E}_{RT}) during water electrolysis and fuel-cell operation using the ATO support were calculated with the following equation at different current densities [52-54]:

$$
\varepsilon_{RT} = \frac{E_{FC}}{E_{WE}},\tag{1}
$$

where E_{FC} and E_{WE} are the cell voltages of the fuel cell and water electrolyser, respectively, at a given current density. The values of round-trip energy efficiency were approximately 48%, 43% and 31% at 50, 100 and 200 A $\rm g^{-1}$, respectively.

4. Conclusions

A support based on Sb-doped $SnO₂$ (ATO) for PEM-URFC has shown promising properties in the PEMWE mode in comparison with an Ebonex $^{\circledast}$ commercial support. The crystal size obtained through this synthesis procedure was $4-6$ nm. The surface values estimated by means of the BET method for ATO are significantly higher than the values obtained for commercial ATO and Ebonex®. Furthermore, the maximum mass current activity was 1118 A g^{-1} at 1.8 V in PEMWE mode (Water electrolyser) and 565 A g^{-1} at 0.3 V in PEMFC mode (Fuel cell), both at 80 \degree C. The value for round-trip energy efficiency was approximately 48% at 50 A $\rm g^{-1}$.

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